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Optimization of nonhomogeneous indium-galliumnitride Schottky-barrier thin-film solar cells

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Abstract. A two-dimensional model was developed to simulate the optoelectronic characteristics of indium-gallium-nitride (In_{ξ}Ga_{1- ξ}N), thin-film, Schottky-barrier solar cells. The solar cells comprise a window designed to reduce the reflection of incident light, Schottky-barrier and ohmic front electrodes, an *n*-doped $In_{\xi}Ga_{1-\xi}N$ wafer, and a metallic periodically corrugated backreflector. The ratio of indium to gallium in the wafer varies periodically throughout the thickness of absorbing layer of the solar cell. Thus, the resulting $In_{z}Ga_{1-z}N$ wafer's optical and electrical properties are made to vary periodically. This material nonhomogeneity could be physically achieved by varying the fractional composition of indium and gallium during deposition. Empirical models for indium nitride and gallium nitride were combined using Vegard's law to determine the optical and electrical constitutive properties of the alloy. The nonhomogeneity of the electrical properties of the $In_{\varepsilon}Ga_{1-\varepsilon}N$ aids in the separation of the excited electron-hole pairs, whereas the periodicities of optical properties and the backreflector enable the incident light to couple to multiple guided wave modes. The profile of the resulting chargecarrier-generation rate when the solar cell is illuminated by the AM1.5G spectrum was calculated using the rigorous coupled-wave approach. The steady-state drift-diffusion equations were solved using COMSOL, which employs finite-volume methods, to calculate the current density as a function of the voltage. Midband Shockley-Read-Hall, Auger, and radiative recombination rates were taken to be the dominant methods of recombination. The model was used to study the effects of the solar-cell geometry and the shape of the periodic material nonhomogeneity on efficiency. The solar-cell efficiency was optimized using the differential evolution algorithm. © 2018 Society of Photo-Optical Instrumentation Engineers (SPIE) [DOI: 10.1117/1.JPE.8.034501]

Keywords: thin-film solar cell; Schottky barrier; indium gallium nitride; periodically corrugated backreflector; optical model; electronic model; nonhomogeneous composition.

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1 Introduction

Alloys of indium gallium nitride $(\ln_{\xi}Ga_{1-\xi}N)$ can be tailored to possess a wide range of bandgaps, from 0.70 to 3.42 eV, by varying the relative proportions of indium and gallium through the parameter $\xi \in (0,1)$.¹ Pure indium nitride (i.e., $\xi = 1$) has a bandgap of 0.7 eV,^{2,3} whereas gallium nitride (i.e., $\xi = 0$) has a bandgap of 3.42 eV. It should be noted that $\ln_{\xi}Ga_{1-\xi}N$ with high indium content (i.e., $\xi \gtrsim 0.3$) currently suffers from poor electrical characteristics, background *n*-doping due to Fermi pinning above the conduction-band edge,⁴ and a bandgap that is greater than expected.^{5,6} These problems are exacerbated by *p*-doping of $\ln_{\xi}Ga_{1-\xi}N$.⁷

Solar cells can be designed to use an in-built potential provided by a Schottky-barrier junction, which can occur at a metal/semiconductor interface.^{8,9} By partnering *n*-doped $In_{\xi}Ga_{1-\xi}N$ with a metal possessing a large work function Φ —as opposed to, say, employing the more usual

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p-i-n junction—the problems associated with *p*-doping of the material are avoided. Furthermore, the deposition process is simplified, as only one dopant element is required. Hence, the reduced fabrication costs could offset the lower efficiencies of Schottky-barrier thin-film solar cells.

Theoretical studies,^{7,10} which corroborate an earlier experimental study,¹¹ suggest that $In_{\xi}Ga_{1-\xi}N$ Schottky-barrier solar cells with relatively high efficiency could be designed. Anderson et al.¹² investigated the efficiency of $In_{\xi}Ga_{1-\xi}N$ Schottky-barrier solar cells with periodic variation of the indium-to-gallium ratio. This involved the solution of both the frequency-domain Maxwell postulates in the optical regime and the carrier drift-diffusion equations using the commercial finite-element package COMSOL (V5.2a), in order to simulate the efficiencies of a variety of designs. The efficiency was found to increase significantly on the incorporation of periodic nonhomogeneity with a specific profile.

For the traditional amorphous-silicon *p-i-n*-junction solar cells, the incorporation of a periodically nonhomogeneous intrinsic layer (i.e., *i* layer), along with a metallic periodically corrugated backreflector (PCBR), can improve overall efficiency by up to 17%.¹³ For a Schottkybarrier solar cell made from $In_{\xi}Ga_{1-\xi}N$, the inclusion of these features was shown to increase the total efficiency by up to 26.8%.¹² In neither case, however, was comprehensive optimization of the design parameters conducted. The improvements seen are likely due to the following reasons:

- i. The excitation of guided wave modes, including surface-plasmon-polariton waves^{14–16} and waveguide modes,¹⁷ is made possible by the inclusion of the metallic PCBR.^{18–21} Both types of phenomena intensify the optical electric field inside the photon-absorbing regions of the solar cell, which leads to an increase in the electron–hole pair generation rate.
- ii. The combination of a periodically nonhomogeneous semiconductor and a PCBR enables the excitation of an increased number of guided wave modes.^{20,22,23} More pathways become available for the incident photons to be absorbed, thereby increasing the charge-carrier-generation rate.
- iii. The drift-diffusion equations include terms pertaining to gradients in the electrical constitutive properties of the materials in the solar cell. The material nonhomogeneity will facilitate the separation of electrons and holes, and it may also suppress recombination.^{24,25}

The aim of this paper is to expand on the previous work on $In_{\xi}Ga_{1-\xi}N$ by providing a comprehensive optimization of the device parameters in order to maximize efficiency. The optical calculations were undertaken using the rigorous coupled-wave approach (RCWA),¹⁶ whereas the electrical calculations were undertaken using COMSOL (V5.3a).²⁶ Optical absorption could have been maximized if only optical models had been used, but the missing influence of the varying electrical properties would have made optimization of efficiency impossible.¹³ For example, if electrical modeling is omitted, the optical absorption can be maximized by minimizing the bandgap, but this would result in a solar cell with a small open-circuit voltage and therefore, quite likely, low efficiency.²⁷

The plan of this paper is as follows. The design of the chosen solar cell is summarized in Sec. 2.1, with further details available elsewhere.^{13,28} The optical and electrical constitutive properties used in the simulation are presented in Sec. 2.2, and the computational models employed are described in Sec. 2.3. Numerical results are presented in Sec. 3. Closing remarks are presented in Sec. 4.

2 Summary of the Two-Dimensional (2-D) Model

2.1 Solar-Cell Design

The model is described in detail in Ref. 12. For the sake of completeness, a summary is included here. The simulated Schottky-barrier solar cell is schematically shown in Fig. 1. As the solar cell is translationally invariant in the y direction, the simulation is reduced to two dimensions (i.e., the xz plane) without approximation. In the remainder of this paper, the term width refers to the extent along the x axis, whereas the term thickness refers to the extent along the z axis.



Fig. 1 Schematic illustration of the reference unit cell \mathcal{R} of the Schottky-barrier solar cell.

The solar cell comprises a planar antireflection window, a layer containing electrodes, a wafer of $In_{\xi}Ga_{1-\xi}N$, and a layer containing a backreflector. Each of these layers is of uniform thickness. Insolation occurs at normal incidence to the solar cell through the antireflection window, with the wave vector of the incident light aligned with the positive *z* axis.

The device is periodic along the x axis with period L_x and has a thickness $L_w + L_c + L_z + L_r$. The reference unit cell of the device is the region $\mathcal{R} = \{(x, z) | -L_x/2 < x < L_x/2, -L_w - L_c < z < L_z + L_r\}$. A planar antireflection window, made from flint glass,²⁹ occupies the region $-L_w - L_c < z < -L_c$ in \mathcal{R} . The region $0 < z < L_z$ is occupied by *n*-doped $\ln_{\xi}Ga_{1-\xi}N$, forming both Schottky-barrier and ohmic junctions with the metal electrodes in the region $-L_c < z < 0$ in \mathcal{R} . For optical calculations, the ohmic contact and backreflector were assumed to be silver,³⁰ whereas the Schottky-barrier contact was assumed to be platinum.³¹ It must be noted that the electrical properties of silver were not used. The Schottky-barrier electrode of width L_s is centered in \mathcal{R} at x = 0 along the x axis. The two ohmic electrodes, each of width $L_o/2$, are centered at $x = \pm (L_x - L_o/2)/2$ in \mathcal{R} . Note that $L_o + L_s < L_x$ and so the electrodes are electrically isolated. It should also be noted that, due to the periodicity of the design, there are an equal number of ohmic and Schottky-barrier electrodes in the solar cell. The gaps between the electrodes, $-L_c < z < 0$ and either $-(L_x - L_o)/2 < x < (L_x - L_o)/2$, are occupied by flint glass.

The region $L_z < z < L_z + L_r$ in \mathcal{R} contains both silver and flint glass. The backreflector is made of two silver slabs welded together. The first slab is optically thick and occupies the region $L_z + L_r - L_m < z < L_z + L_r$. The second slab occupies the region $\{(x, z) | -\zeta L_x/2 < x < \zeta L_x/2, L_z + L_d < z < L_z + L_r - L_m\} \subset \mathcal{R}$, where $\zeta \in (0,1)$ is the duty cycle. Thus, $L_g = L_r - (L_d + L_m)$ is the corrugation height. The remainder of the region $L_z < z < L_z + L_r$ in \mathcal{R} is occupied by flint glass which electrically insulates silver from $\ln_{\xi} Ga_{1-\xi}N$.

Absorption of the normally incident solar flux with AM1.5G spectrum³² was calculated by solving the frequency-domain Maxwell postulates.¹⁶ The semiconductor charge-carrier drift-diffusion equations model the spatial distributions of the electron density and the hole density.^{33,34} Because of the nonhomogeneity of the semiconductor (i.e., $In_{\xi}Ga_{1-\xi}N$), the effective dc electric field acting on

- a. electrons includes a contribution from gradients in the electron affinity, and
- b. holes includes contributions from gradients in both the electron affinity and the bandgap.

Direct, midgap Shockley–Read–Hall, and Auger recombination were all included in our simulation. The current density J, which is averaged over either the Schottky-barrier electrode (or, identically, both of the ohmic electrodes), was calculated for a range of values of the external biasing voltage V_{ext} .

2.2 Material Parameters

For a specific bandgap E_{q0} , the fractional concentration of indium ξ is given as

$$\xi(z) = \frac{b + (\mathsf{E}_{g}^{\text{GaN}} - \mathsf{E}_{g}^{\text{InN}}) - \sqrt{4b[\mathsf{E}_{g0}(z) - \mathsf{E}_{g}^{\text{GaN}}] + (b + \mathsf{E}_{g}^{\text{GaN}} - \mathsf{E}_{g}^{\text{InN}})^{2}}}{2b},$$
(1)

where the bowing parameter b = 1.43 eV,^{12,35} and the bandgaps $E_g^{\text{InN}} = 0.7$ eV of InN and $E_g^{\text{GaN}} = 3.42$ eV of GaN.

2.2.1 Optical parameters

The optical refractive index n_{opt} of $In_{\xi}Ga_{1-\xi}N$ depends on the free-space wavelength λ_0 and was modeled using two equations. The real part of n_{opt} is provided by the Adachi model as⁷

$$\operatorname{Re}\{n_{\operatorname{opt}}(\xi,\lambda_{0})\} = \operatorname{Re}\left(\sqrt{A_{A}(\xi)}\left\{\left[\frac{\mathsf{E}_{g0}(\xi)}{\mathsf{E}_{\gamma}(\lambda_{0})}\right]^{2}\left[2 - \sqrt{1 + \frac{\mathsf{E}_{\gamma}(\lambda_{0})}{\mathsf{E}_{g0}(\xi)}} - \sqrt{1 - \frac{\mathsf{E}_{\gamma}(\lambda_{0})}{\mathsf{E}_{g0}(\xi)}}\right] + B_{A}(\xi)\right\}\right),\tag{2}$$

where $A_A(\xi)$ and $B_A(\xi)$ are interpolated from the corresponding parameters for InN and GaN provided in Table 1. The photon energy is denoted by $E_{\gamma}(\lambda_0) = 2\pi \hbar c_0/\lambda_0$, where $\hbar = 1.054571800 \times 10^{-34} \text{ m}^2 \text{ kg s}^{-1}$ is the reduced Planck constant and $c_0 = 2.99792485 \times 10^8 \text{ m s}^{-1}$ is the speed of light in free space.

The imaginary part of the optical refractive index n_{opt} was modeled as

$$\operatorname{Im}\{n_{\rm opt}(\xi,\lambda_0)\} = \frac{\lambda_0}{4\pi} \alpha_{\rm opt}(\xi,\lambda_0).$$
(3)

The absorption coefficient α_{opt} was modeled as⁷

$$\alpha_{\rm opt}(\xi,\lambda_0) = 10^5 \sqrt{C(\xi) [\mathsf{E}_{\gamma}(\xi) - \mathsf{E}_{g0}(\xi)] + D(\xi) [\mathsf{E}_{\gamma}(\xi) - \mathsf{E}_{g0}(\xi)]^2} \,\,\mathrm{nm^{-1}},\tag{4}$$

wherein the constants

$$C(\xi) = (3.525 - 18.28\xi + 40.22\xi^2 - 37.52\xi^3 + 12.77\xi^4) \text{ eV}^{-1} \},$$

$$D(\xi) = (-0.6651 + 3.616\xi - 2.460\xi^2) \text{ eV}^{-2} \},$$
(5)

come from interpolation of parameters given by Brown et al.³⁶

2.2.2 Electrical parameters

The Schottky-barrier work function matched that of platinum in our simulations. Thus, $\Phi = 5.93 \text{ eV.}^{37}$ For a specific value of ξ , the electrical properties were modeled using either quadratic or linear (i.e., Vegard's law³⁸) interpolation of data for InN and GaN.

Table 1	Electronic data	used for GaN a	nd InN. The	e compo	sition of	f In _č Ga₁	<i>_ξ</i> N was	estimate	۶d
using Eq.	(1), with linear	interpolation use	ed to estima	te data	for the	semicor	nductor-fil	led regio	n
$0 < z < L_{2}$	z with bandgaps	s not presented	here in all	cases,	except	for the	electron	affinity ;	łо
which use	es Eq. (<mark>6</mark>).								

	Symbol	Unit	GaN	InN
Bandgap	E_g^*	eV	3.42	0.7
Electron affinity	χo	eV	4.1	5.6
Density of states (conduction band)	N _C	cm ⁻³	2.3×10^{18}	9.1×10^{17}
Density of states (valence band)	N_V	cm ⁻³	4.6×10^{19}	$5.3 imes10^{19}$
Electron mobility 1	$\mu_n^{(1)}$	$cm^2 V^{-1} s^{-1}$	295	1030
Electron mobility 2	$\mu_n^{(2)}$	${\rm cm}^2 {\rm V}^{-1} {\rm s}^{-1}$	1460	14150
Caughey-Thomas doping power (electrons)	δ_n		0.71	0.6959
Caughey-Thomas critical doping density (electrons)	N ^{crit}	cm ⁻³	7.7×10^{16}	2.07×10^{16}
Hole mobility 1	$\mu_p^{(1)}$	$cm^2 V^{-1} s^{-1}$	3	3
Hole mobility 2	$\mu_p^{(2)}$	$cm^2 V^{-1} s^{-1}$	170	340
Caughey-Thomas doping power (holes)	δ_p		2	2
Caughey-Thomas critical doping density (holes)	$N_p^{\rm crit}$	cm ⁻³	1×10^{18}	8×10^{17}
Auger recombination factor (electrons)	C _n	cm ⁶ s ⁻¹	$1.5 imes 10^{-30}$	$1.5 imes 10^{-30}$
Auger recombination factor (holes)	C_p	cm ⁶ s ⁻¹	$1.5 imes 10^{-30}$	$1.5 imes 10^{-30}$
Direct recombination factor	\mathcal{C}_{rad}	$\mathrm{cm}^3\mathrm{s}^{-1}$	1.1 × 10 ⁻⁸	2×10^{-10}
Slotboom reference energy	$E_{\rm ref}$	eV	9×10^{-3}	9 × 10 ⁻³
Slotboom reference concentration	N _{ref}	cm ⁻³	1×10^{17}	1×10^{17}
Conduction-band fraction	$\alpha_{ m oc}$		0.9	0.9
Adachi refractive-index parameter A_A	A _A		9.31	13.55
Adachi refractive-index parameter B_A	B _A		3.03	2.05

The electron affinity

$$\chi_0(z) = \xi(z)\chi^{\text{InN}} + [1 - \xi(z)]\chi^{\text{GaN}} - b\xi(z)[1 - \xi(z)], \qquad z \in (0, L_z),$$
(6)

was modeled using the same quadratic fit as the bandgap, where χ^{InN} and χ^{GaN} are the electron affinities of InN and GaN, respectively. All other parameters presented in the first column of Table 1 were modeled using Vegard's law of linear interpolation.

The narrowing of the bandgap associated with doping was incorporated through the Slotboom model.³⁹ An empirical low-field mobility model—called either the Caughey–Thomas⁷ or the Arora²⁶ mobility model—was used for the variations of the electron mobility and the hole mobility. Details of these models are available elsewhere.^{7,12}

The bandgap of $In_{\xi}Ga_{1-\xi}N$ was taken to vary periodically in the thickness direction of the solar cell, described as

$$\mathsf{E}_{g0}(z) = \mathsf{E}_g^* - A\left(1 - \left\{\frac{1}{2}\left[\sin\left(\frac{2\pi z}{L_p} - 2\pi\phi\right) + 1\right]\right\}^{\alpha}\right),\tag{7}$$

where E_g^* is the baseline (maximum) bandgap, A is the amplitude, $L_p = L_z/\tilde{\kappa}$ is the period with the bandgap nonhomogeneity ratio $\tilde{\kappa} > 0$, ϕ is a phase shift, and α is a shaping parameter.

2.3 Computational Model

The problem of calculating the total efficiency of the solar cell was decoupled into two separate calculations. First, the RCWA¹⁶ was used to calculate the spectrally integrated photon-absorption rate, which is ideally equal to the charge-carrier-generation rate. This was then coupled with a 2-D finite-element electronic model that was implemented in the COMSOL (V5.3a) software package.²⁶ In the remainder of this paper, terms in small capitals are COMSOL terms.

2.3.1 Differential evolution algorithm

The differential evolution algorithm (DEA)⁴⁰ was used to optimize the solar-cell design. Given \overline{N} parameters in the optimization problem, an initial population \mathbf{P}_0 of N_P members in the parameter-search space $S \subset \mathbb{R}^{\overline{N}}$ was chosen randomly in [0.5,1] with a uniform distribution. After the cost function $C: S \to \mathbb{R}$ of the problem had been evaluated at each of these points, the DEA produced a new population \mathbf{P}_1 of N_P members in the parameter search space S to test. This process was iterated until the change in absorptance was less than 1% or until a set amount of time had passed.

By representing the current population \mathbf{P}_j as a matrix with each of its N_P columns being vectors in S, the next population can be written as

$$\mathbf{P}_{j+1} = \mathbf{M}_1 \circ [\mathbf{P}_j + F(\mathbf{v}^* \otimes \mathbf{1} - \mathbf{P}_j) + F(\tilde{\mathbf{P}}_j^{(1)} - \tilde{\mathbf{P}}_j^{(2)})] + \mathbf{M}_2 \circ \mathbf{P}_j,$$
(8)

where $\mathbf{v}^* \in S$ is the optimal parameter vector found at that stage; **1** is the vector of 1's; \otimes is the outer product; $\tilde{\mathbf{P}}_j^{(1)}$ and $\tilde{\mathbf{P}}_j^{(2)}$ are versions of \mathbf{P}_j where the columns have been randomly interchanged; the parameter $F \in (0,2]$ is the step size to be taken by the DEA at each iteration; \mathbf{M}_1 and $\mathbf{M}_2 = \mathbf{1} \otimes \mathbf{1} - \mathbf{M}_1$ are filter matrices of 1's and 0's generated by the DEA, with \mathbf{M}_1 having approximately a fraction C_R of 1's, where C_R is termed the crossover fraction; and \circ is the Shur product (elementwise multiplication).

The cost function C was taken to be either the efficiency η or the optical short-circuit current density J_{SC}^{Opt} . The population number was set to $N_P = 30$, the crossover fraction was set to $C_R = 0.6$, and the step size F was set to be randomly distributed in [0.5, 1] uniformly. Allowing F to vary randomly with each iteration has been termed dither, and has been shown to improve convergence for many problems.⁴¹

2.3.2 RCWA algorithm

Suppose that the face $z = -L_c - L_w$ of the solar cell is illuminated by a normally incident plane wave with electric field phasor

$$\underline{E}_{\rm inc}(x, z, \lambda_0) = \frac{E_o}{\sqrt{2}} (\hat{\underline{x}} + \hat{\underline{y}}) \exp\left(i\frac{2\pi}{\lambda_0}z\right).$$
(9)

As a result of the metallic backreflector being periodically corrugated, the *x*-dependences of the electric and magnetic field phasors must be represented by Fourier series everywhere as

$$\underline{E}(x,z,\lambda_0) = \sum_{n=-\infty}^{n=\infty} \underline{e}^{(n)}(z,\lambda_0) \exp(i\kappa^{(n)}x), \quad |z| < \infty, \quad |x| < \infty,$$
(10)

and

$$\underline{H}(x,z,\lambda_0) = \sum_{n=-\infty}^{n=\infty} \underline{h}^{(n)}(z,\lambda_0) \exp(i\kappa^{(n)}x), \quad |z| < \infty, \quad |x| < \infty,$$
(11)

where $i = \sqrt{-1}$, $\kappa^{(n)} = n(\frac{2\pi}{L_x})$, and $\underline{e}^{(n)} = e_x^{(n)}\hat{\underline{x}} + e_y^{(n)}\hat{\underline{y}} + e_z^{(n)}\hat{\underline{z}}$ as well as $\underline{h}^{(n)} = h_x^{(n)}\hat{\underline{x}} + h_y^{(n)}\hat{\underline{y}} + h_z^{(n)}\hat{\underline{z}}$ are Fourier coefficients. Likewise, the optical permittivity $\varepsilon(x, z, \lambda_0) = \varepsilon_0 n_{opt}^2(x, z, \lambda_0)$ everywhere has to be represented by the Fourier series

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$$\varepsilon(x, z, \lambda_0) = \sum_{n=-\infty}^{n=\infty} \varepsilon^{(n)}(z, \lambda_0) \exp(i\kappa^{(n)}x), \quad |z| < \infty, \quad |x| < \infty,$$
(12)

where ε_0 is the permittivity of free space.

Computational tractability requires truncation so that $n \in \{-N_t, ..., N_t\}, N_t \ge 0$. Column vectors

$$\check{\mathbf{e}}_{\sigma}(z,\lambda_0) = \left[e_{\sigma}^{(-N_t)}(z,\lambda_0), e_{\sigma}^{(-N_t+1)}(z,\lambda_0), \dots, e_{\sigma}^{(N_t-1)}(z,\lambda_0), e_{\sigma}^{(N_t)}(z,\lambda_0)\right]^T, \qquad \sigma \in \{x, y, z\},$$
(13)

and

$$\check{\mathbf{h}}_{\sigma}(z,\lambda_{0}) = [h_{\sigma}^{(-N_{t})}(z,\lambda_{0}), h_{\sigma}^{(-N_{t}+1)}(z,\lambda_{0}), \dots, h_{\sigma}^{(N_{t}-1)}(z,\lambda_{0}), h_{\sigma}^{(N_{t})}(z,\lambda_{0})]^{T}, \qquad \sigma \in \{x, y, z\},$$
(14)

were set up, the superscript T denoting the transpose. Furthermore, the matrices

$$\mathbf{\check{K}} = \operatorname{diag}[\boldsymbol{\kappa}^{(-N_t)}, \boldsymbol{\kappa}^{(-N_t+1)}, \dots, \boldsymbol{\kappa}^{(N_t-1)}, \boldsymbol{\kappa}^{(N_t)}]$$
(15)

and

$$\breve{\boldsymbol{\varepsilon}}(z,\lambda_{0}) = \begin{bmatrix}
\varepsilon^{(0)}(z,\lambda_{0}) & \varepsilon^{(-1)}(z,\lambda_{0}) & \dots & \varepsilon^{(-2N_{t}+1)}(z,\lambda_{0}) & \varepsilon^{(-2N_{t})}(z,\lambda_{0}) \\
\varepsilon^{(1)}(z,\lambda_{0}) & \varepsilon^{(0)}(z,\lambda_{0}) & \dots & \varepsilon^{(-2N_{t}+2)}(z,\lambda_{0}) & \varepsilon^{(-2N_{t}+1)}(z,\lambda_{0}) \\
\dots & \dots & \dots & \dots & \dots \\
\varepsilon^{(2N_{t}-1)}(z,\lambda_{0}) & \varepsilon^{(2N_{t}-2,\lambda_{0})}(z,\lambda_{0}) & \dots & \varepsilon^{(0)}(z,\lambda_{0}) & \varepsilon^{(-1)}(z,\lambda_{0}) \\
\varepsilon^{(2N_{t})}(z,\lambda_{0}) & \varepsilon^{(2N_{t}-1)}(z,\lambda_{0}) & \dots & \varepsilon^{(1)}(z,\lambda_{0}) & \varepsilon^{(0)}(z,\lambda_{0})
\end{bmatrix} (16)$$

were set up. The frequency-domain Maxwell curl postulates then yielded the matrix ordinary differential equation

$$\frac{d}{dz}\check{\mathbf{f}}(z,\lambda_0) = i\check{\mathbf{P}}(z,\lambda_0) \bullet \check{\mathbf{f}}(z,\lambda_0),$$
(17)

where the $4(2N_t + 1)$ -column vector

$$\check{\mathbf{f}}(z,\lambda_0) = \begin{bmatrix} \check{\mathbf{e}}_x(z,\lambda_0) \\ \check{\mathbf{e}}_y(z,\lambda_0) \\ \check{\mathbf{h}}_x(z,\lambda_0) \\ \check{\mathbf{h}}_y(z,\lambda_0) \end{bmatrix}$$
(18)

and the $4(2N_t+1) \times 4(2N_t+1)$ matrix

$$\mathbf{\check{P}}(z,\lambda_0) = \omega \begin{bmatrix} \check{\mathbf{0}} & \check{\mathbf{0}} & \check{\mathbf{0}} & \mu_0\check{\mathbf{I}} \\ \check{\mathbf{0}} & \check{\mathbf{0}} & -\mu_0\check{\mathbf{I}} & \check{\mathbf{0}} \\ \check{\mathbf{0}} & -\check{\boldsymbol{\varepsilon}}(z,\lambda_0) & \check{\mathbf{0}} & \check{\mathbf{0}} \\ \check{\boldsymbol{\varepsilon}}(z,\lambda_0) & \check{\mathbf{0}} & \check{\mathbf{0}} & \check{\mathbf{0}} \end{bmatrix} \\
+ \frac{1}{\omega} \begin{bmatrix} \check{\mathbf{0}} & \check{\mathbf{0}} & \check{\mathbf{0}} & -\check{\mathbf{K}} \cdot [\check{\boldsymbol{\varepsilon}}(z,\lambda_0)]^{-1} \bullet \check{\mathbf{K}} \\ \check{\mathbf{0}} & \check{\mathbf{0}} & \check{\mathbf{0}} & \check{\mathbf{0}} \\ \check{\mathbf{0}} & \mu_0^{-1}\check{\mathbf{K}} \bullet \check{\mathbf{K}} & \check{\mathbf{0}} & \check{\mathbf{0}} \\ \check{\mathbf{0}} & \check{\mathbf{0}} & \check{\mathbf{0}} & \check{\mathbf{0}} \end{bmatrix} \tag{19}$$

contains μ_0 as the permeability of free space, $\check{\mathbf{0}}$ as the $(2N_t + 1) \times (2N_t + 1)$ null matrix, and $\check{\mathbf{I}}$ as the $(2N_t + 1) \times (2N_t + 1)$ identity matrix.

The solar cell was discretized along the z axis.¹⁶ This effectively broke the domain \mathcal{R} into a cascade of slices. Each slice was homogeneous along the z axis, but it was either homogeneous or periodically nonhomogeneous along the x axis. Equation (17) was then solved using

a stepping algorithm to give an approximation for $\mathbf{\check{f}}$ in each slice. Finally, the Fourier coefficients of the *z* components of the electric and magnetic field phasors were obtained from

$$\begin{split} \check{\mathbf{e}}_{z}(z,\lambda_{0}) &= -\omega^{-1} [\check{\boldsymbol{e}}(z,\lambda_{0})]^{-1} \bullet \check{\mathbf{K}} \bullet \check{\mathbf{h}}_{y}(z,\lambda_{0}) \\ \check{\mathbf{h}}_{z}(z,\lambda_{0}) &= (\omega\mu_{0})^{-1} \check{\mathbf{K}} \bullet \check{\mathbf{e}}_{y}(z,\lambda_{0}) \end{split}$$

$$\end{split}$$

$$(20)$$

Thus, the electric field phasor was determined throughout the solar cell.

The spectrally integrated number of absorbed photons per unit volume per unit time is given as

$$N_{\rm ph}(x,z) = \frac{\eta_0}{\hbar} \int_{\lambda_{0_{\rm min}}}^{\lambda_{0_{\rm max}}} \operatorname{Im}\{\varepsilon(x,z,\lambda_0)\} \left| \frac{\underline{E}(x,z,\lambda_0)}{E_o} \right|^2 S(\lambda_0) d\lambda_0,$$
(21)

where $\eta_0 = \sqrt{\mu_0/\varepsilon_0}$ is the intrinsic impedance of free space and $S(\lambda_0)$ is the AM1.5G solar spectrum.³² With the assumption that the absorption of every photon in the $\ln_{\xi}Ga_{1-\xi}N$ layer releases an electron–hole pair, the charge-carrier-generation rate can be calculated as

$$G(x, z) = N_{\rm ph}(x, z), \qquad (22)$$

everywhere in that layer. Whereas $\lambda_{0_{\min}} = 350 \text{ nm}$, $\lambda_{0_{\max}} = 1240 \text{ nm eV/E}_{g,\min}$ is the maximum wavelength that can contribute to the optical short-circuit current density

$$J_{\rm SC}^{\rm Opt} = q_e \frac{1}{L_x} \int_0^{L_z} \int_{-L_x/2}^{L_x/2} G(x, z) dx dz,$$
(23)

where $E_{g,\min}$ (in eV) is the minimum bandgap present in the solar cell and $q_e = 1.6 \times 10^{-19}$ C is the elementary charge.

The integral in Eq. (21) was approximated using the trapezoidal rule⁴³ with sampling at wavelengths spaced at 2-nm intervals. The integral in Eq. (23) was also approximated using the trapezoidal rule. The sampling resolution was regular in both directions, with $\delta x = L_x/100$, and $\delta z = L_z/200$.

The optical short-circuit current density provides a rough benchmark for the device efficiency and is used by many optics researchers who simulate solar cells.⁴² However, as recombination is neglected, J_{SC}^{Opt} is necessarily larger than the actually attainable short-circuit current density J_{SC} , which is the electronically simulated current density that flows when the solar cell is illuminated and no external bias is applied (i.e., when $V_{ext} = 0$). For the results presented here, calculating only J_{SC}^{Opt} would have been inadequate as the electrical constitutive properties were also significantly varied.

2.3.3 Adaptive-N_t implementation

The calculated value of J_{SC}^{Opt} varies with N_t . An adaptive method was implemented to estimate when N_t is sufficiently large. Equation (21) was evaluated using the trapezoidal rule.⁴³ At the first value of λ_0 sampled, $\lambda_0 = \lambda_{0_{min}}$, J_{SC}^{Opt} was calculated with N_{t_1} and $N_{t_2} = N_{t_1} + 2$. If the magnitude $\Delta J_{SC}^{Opt} = |J_{SC}^{Opt}(N_{t_2}) - J_{SC}^{Opt}(N_{t_1})|$ of the difference between the two calculated values of J_{SC}^{Opt} was greater than a specified tolerance, then N_{t_1} was set equal to N_{t_2} and N_{t_2} was increased by two. This iterative procedure was continued until ΔJ_{SC}^{Opt} was less than the specified tolerance for two subsequent comparisons. A maximum value of $N_t \leq 100$ was enforced in order to force to the calculation to terminate within a reasonable duration. After a successful calculation, the next value of λ_0 was selected, and $N_{t_1} = 2\lceil N_{t_2}/4\rceil$ and $N_{t_2} = N_{t_1} + 2$ were chosen for the next calculation, where $\lceil \rceil$ is the ceiling function.

2.3.4 Solution of drift-diffusion equations

The charge-carrier-generation rate G(x, z), calculated using the RCWA, was processed using an external MatlabTM code and then used as the input, via USER-DEFINED GENERATION, for the COMSOL electrical model. Recombination was incorporated via AUGER, DIRECT and TRAP-ASSISTED (MIDGAP SHOCKLEY-READ-HALL) phenomena, using parameters as provided in Table 1.

Due to the symmetry in the simulation, only the right half of the domain \mathcal{R} (i.e., $0 \le x \le L_x/2$) was electrically simulated, with INSULATOR INTERFACES applied down the lines of symmetry.

FERMI–DIRAC CARRIER STATISTICS were employed along with a FINITE VOLUME (CONSTANT SHAPE FUNCTION) discretization, as this inherently conserves current throughout the solar cell.²⁶ COMSOL utilizes a Scharfetter–Gummel upwinding scheme. The FREE TRIANGULAR, DELAUNAY mesh has a maximum element size of 15 nm. Further details can be found in Ref. 12.

The semiconductor module of COMSOL was used to calculate the current densities flowing through the OHMIC, and therefore also IDEAL SCHOTTKY, electrodes. A prescribed external voltage V_{ext} was applied between these electrodes. The current density flowing through the Schottky-barrier electrode was modeled using THERMIONIC CURRENTS, with standard Richardson coefficients of $A_n^* = 110 \text{ A K}^{-2} \text{ cm}^{-2}$ and $A_p^* = 90 \text{ A K}^{-2} \text{ cm}^{-2}$.^{7,26} By sweeping V_{ext} from 0 V up to a value where J_{SC} drops to zero, the J_{SC} - V_{ext} curve was produced. This enabled calculation of the maximum attainable value of the efficiency η .

3 Numerical Simulation Results

3.1 Optimization Study

The defined problem has 15 parameters, shown in Table 2, all of which influence the chargecarrier-generation rate G(x, z) and the efficiency η of the solar cell. The choice of four of these parameters was guided by either physical constraints or they were found to strongly affect the optical response of the solar cell while affecting the electrical characteristics only indirectly, i.e., by changing the spatial profile of the charge-carrier-generation rate.

- A smaller value of L_d was seen to further increase the number of photons absorbed in the $\ln_{\xi}Ga_{1-\xi}N$; however, a minimum value of $L_d = 40$ nm was chosen: (i) to model a solar cell where the electrical insulation of the backreflector is maintained, and (ii) to provide a suitable surface for $\ln_{\xi}Ga_{1-\xi}N$ deposition.
- The effect of the minimum thickness L_m of the metallic layer has been omitted from the results. This is because, given a sufficiently thick silver layer, only a trivial amount of the incident light will be transmitted through the solar cell. With L_m chosen to be more than twice the penetration depth of silver across the majority of the AM1.5G spectrum, changes in solar-cell performance for modest perturbations in L_m from this value are minimal.
- A smaller value of L_c was seen to further increase the number of photons absorbed in the $In_{\xi}Ga_{1-\xi}N$. A minimum value of $L_c = 50$ nm was chosen so the contacts are more physically realistic: thinner contacts would likely suffer from uneven deposition and exhibit increased series resistance.
- A strong peak in efficiency was seen when $L_w \approx 75$ nm, irrespective of the other parameters. This corresponds to the glass coating acting as a quarter-wavelength antireflection coating.

For all data reported here, the values $L_d = 50$ nm, $L_m = 100$ nm, $L_c = 50$ nm, and $L_w = 75$ nm were fixed.

The remaining 11 parameters were allowed to vary within the following ranges: $L_x \in [100, 1500] \text{ nm}, L_z \in [20, 800] \text{ nm}, L_g \in [1, 200] \text{ nm}, \zeta \in [0.01, 0.99], L_o \in [50, L_x - L_s] \text{ nm}, L_s \in [50, L_x - L_o] \text{ nm}, \mathbb{E}_g^* \in [0.7, 3] \text{ eV}, A \in [0, 3.42 - \mathbb{E}_g^*] \text{ eV}, \phi \in [0, 1), \alpha \in [2^{-4}, 2^8],$ and $\tilde{\kappa} \in [0.01, 5.5]$. These ranges, along with the chosen values of the optical parameters, are summarized in Table 2.

The maximum obtained efficiency $\eta = 11.13\%$ was found at: $L_x = 694$ nm, $L_g = 125$ nm, $\zeta = 0.012$, $L_o/L_x = 0.16$, $L_s/L_x = 0.01$, A = 0.74025, $\tilde{\kappa} = 3.05$, $\phi = 0.711$, $\alpha = 13.3614$,

Parameter	Name	Symbol	Value/Range
1	Device period	L _x	[100,1500] nm
1	Thickness of <i>n</i> -doped $In_{\xi}Ga_{1-\xi}N$ layer	Lz	{20,800} nm
	Minimum thickness of insulation window	L _d	40 nm
1	Corrugation height	L_g	[1,200] nm
1	Corrugation duty cycle	ζ	[0.01,0.99]
	Minimum thickness of metal	L _m	100 nm
	Electrode-region thickness	L _c	50 nm
1	Ohmic-electrode width	Lo	50+ nm
1	Schottky-barrier electrode width	Ls	50+ nm
	Antireflection-window thickness	L_w	75 nm
1	Pre-doping bandgap	E_g^*	[0.7,3] eV
1	Bandgap nonhomogeneity amplitude	A	$[0, 3.42 - E_g^*] \text{ eV}$
1	Bandgap nonhomogeneity phase	ϕ	[0, 1]
1	Bandgap nonhomogeneity shaping parameter	α	[2 ⁻⁴ , 2 ⁸]
1	Bandgap nonhomogeneity ratio	$\tilde{\kappa}$	[0.01, 5.5]

Table 2 Summary of parameters used for simulation.

 $E_g^* = 1.17$ eV, and $L_z = 735$ nm. These values were computed after 10 DEA population evolutions and provide an estimate for the maximum efficiency, as well as allowing the upcoming interpretation of the results. Further iterations would provide greater confidence in the conclusions, at the cost of greater computation time.

3.1.1 Results of optimization study

Figure 2 shows η plotted against J_{SC}^{Opt} , with each data point corresponding to one DEA population member. The maximum value of J_{SC}^{Opt} was calculated to be 37.042 mA cm⁻², but the maximum efficiency was observed to occur at $J_{SC}^{Opt} = 31.442$ mA cm⁻². Whereas a larger value of J_{SC}^{Opt}



Fig. 2 (a) Comparison of the solar-cell efficiency η to the optical short-circuit current density J_{SC}^{Opt} , showing that an increase in the latter does not necessarily translate into an increase in the former. (b) Relationship between optical short-circuit current density and simulated short-circuit current density. For all data points, $L_d = 50$ nm, $L_m = 100$ nm, $L_c = 50$ nm, and $L_w = 75$ nm were fixed, but the other parameters were varied.



Fig. 3 Results projected onto the plane containing η and (a) L_z , (b) E_g^* , (c) A, (d) $\tilde{\kappa}$, (e) ϕ , (f) α , (g) L_o/L_x , and (h) L_s/L_x for a solar cell with $L_d = 50$ nm, $L_m = 100$ nm, $L_c = 50$ nm, and $L_w = 75$ nm. The large points highlight the location of the device with the maximum efficiency.

increases the likelihood of obtaining a high value of η , the former does not predict the latter. Indeed, the designs with values of $J_{SC}^{Opt} > 25 \text{ mA cm}^{-2}$ produced efficiencies ranging from <1% up to over 11% in Fig. 2(a). In part, this is caused by a device with a large optical short-circuit current density J_{SC}^{Opt} not automatically producing a large short-circuit current density J_{SC} . This behavior is shown by the droop at higher values of J_{SC}^{Opt} in Fig. 2(b): as the optical short-circuit current density increases, recombination in the solar cell increases. These observations highlight the importance of conducting full optoelectronic simulations when modeling solar cells, especially when parameters with a strong electrical effect, such as the bandgap, are allowed to vary.

3.1.2 Details of optimization study

The results from the optimization study show how the different parameters affect the efficiency η of the solar cell. Figures 3 and 4 show the projections of the entire parameter space onto the sets of axes containing the efficiency and each of the optimization parameters. Parameters that have a strong effect on the efficiency have most of their points strongly clustered around the design with the highest efficiency. Each point is colored with the value of J_{sc}^{Opt} .

In Fig. 3(a), the thickness of the solar cell is seen to have a moderate effect on the resulting efficiency. The peak visible around $L_z \approx 700$ nm sees light clustering, with some reasonably efficient solar cells with $\eta > 7\%$, also produced when L_z is less than half the optimal value. Figure 3(b) shows that E_g^* strongly affects the resulting solar cell efficiency. The peak around 1.2 eV lies in the region predicted by the Shockley–Queisser limit. Solar cells with narrower bandgaps do produce solar cells with higher optical short-circuit current densities, but the reduction in open-circuit voltage dramatically reduces the efficiency. Figure 3(c) shows that a nonzero amplitude A can substantially increase the efficiency. While the structure of the peak is not well resolved, all solar cells with A < 0.5 eV had efficiencies less than half that of the maximum attained efficiency. Further increase of A beyond 1 eV slowly reduces the



Fig. 4 (a) Bandgap of most efficient design. Rest, as Fig. 3 except that the results are projected onto the plane containing η and (b) L_x , (c) L_g , and (d) ζ .

attainable efficiency. Figures 3(g) and 3(h) show that a small Schottky-barrier electrode and a slightly larger ohmic electrode are required to maximize the efficiency.

A previous study¹² has suggested that optimal values of $\tilde{\kappa}$ are integer multiples of 1.5, which conclusion is not contradicted by the data; see Fig. 3(d). The strong peak around $\phi = 0.75$ in Fig. 3(e) is also in line with previous work.^{12,13} At this value of ϕ , a wide bandgap is produced near to the electrodes. This seems of paramount importance for producing high efficiency solar cells with nonhomogeneous bandgaps. The best bandgap profile is shown in Fig. 4(a).

Finally, in Fig. 4, the effects of the PCBR are shown. A period of $L_x > 500$ nm is seen to significantly increase solar cell efficiency. This because the majority of short-wavelength light is



Fig. 5 Properties of the optimal solar cell at short-circuit condition: (a) generation rate in $\log_{10}(\text{cm}^{-3} \text{ s}^{-1})$, (b) recombination rate in $\log_{10}(\text{cm}^{-3} \text{ s}^{-1})$, (c) current density in $\log_{10}(\text{mA} \text{ cm}^{-3})$, (d) current density in *z*-direction in $\log_{10}(\text{mA} \text{ cm}^{-3})$, (e) electron density in $\log_{10}(\text{cm}^{-3})$, and (f) hole density in $\log_{10}(\text{cm}^{-3})$.



Fig. 6 The resulting JV curve for the most efficient solar cell design.

absorbed far from the PCBR and so scattering effects are minimal. The grating amplitude and duty cycle are not seen to have strong effects on the solar-cell efficiency—the points do not cluster strongly in Figs. 4(c) and 4(d).

3.2 Detailed Study

The simulated values of the major variables for the highest-efficiency device are shown in Fig. 5. The first two subfigures, 5(a) and 5(b), show the generation rate and the recombination rate within the solar cell when the external voltage is zero. Light is incident from below in Fig. 5. The bands of low generation and low recombination at $z \approx \{230,570,710\}$ nm correspond to the locations where the bandgap perturation is large, i.e., the bandgap is much wider here. The majority of the photons absorbed in the first 200 nm of the solar cell are collected before recombining, whereas the majority of those absorbed in the rear 500 nm recombine. The exception to this trend are the photons that are absorbed where the bandgap peaks. These are quickly swept out of these regions by the effective electric field produced by the gradient in the bandgap and electron affinity.

Figures 5(c) and 5(d) show the current density produced in the solar cell. By comparing the two figures, it is seen that in the region of the narrow ohmic contact, at the outer edges of the plots, the current density is strongly perpendicular to the contact. The current density in the vicinity of the Schottky barrier (at the center of the plots) is lower, but it is also perpendicular to the contact. The current density toward the back of the solar cell is dramatically lower, supporting the earlier analysis that the majority of the excited carriers in this region recombine.

Figures 5(e) and 5(f) show the electron and hole densities at the short-circuit condition. The electrons are the majority carriers in the majority of the solar cell. In the vicinity of the Schottky barrier, there is a high concentration of holes. In the regions where the bandgap is large, both carrier densities are very low, which has the effect of drastically reducing recombination in these areas. Unfortunately, it also acts to limit current flow across these bands: it may therefore be beneficial in future studies to include lower-bandgap pathways to increase charge-carrier extraction from further back in the solar cell.

Finally, Fig. 6 shows the resulting JV curve for the optimal device. The short-circuit current is 21.79 mA cm^{-2} , the open-circuit voltage is 0.683 V, and the fill factor is 0.7479.

4 Closing Remarks

A combined optoelectronic model was developed to enable to optimization of $In_{\xi}Ga_{1-\xi}N$ -based Schottky-barrier solar cells. These solar cells possessed a PCBR, a layer of $In_{\xi}Ga_{1-\xi}N$, metallic electrodes, and an antireflection coating. The optimization was conducted using the DEA. The AM1.5G solar spectrum was used to illuminate the solar cell at normal incidence. With a solely optical model, it was shown that the optical short-circuit current density J_{SC}^{Opt} of the design is strongly dependent on the thicknesses of the materials that are applied to the surface. A 75-nm thick layer of flint glass acts as a quarter-wavelength antireflection coating,^{44,45} maximizing J_{SC}^{Opt} . Minimizing the thickness of the front metallic electrodes also maximizes J_{SC}^{Opt} by reducing reflection.

An optimization study, using the DEA and a full optoelectronic model, produced a design for an $In_{\xi}Ga_{1-\xi}N$ -based Schottky-barrier solar cell with a simulated efficiency of 11.13%. This design included a periodically nonhomogeneous bandgap, with just over three full periods. The minimum bandgap was 1.17 eV and the maximum bandgap was 1.91 eV. The phase of the periodic nonhomogeneity was such that the $In_{\xi}Ga_{1-\xi}N$ close to the electrodes has a wide bandgap.

While experimental work is needed to test the veracity of the models employed, it has been shown that $In_{\xi}Ga_{1-\xi}N$ -based Schottky-barrier solar cells with a high efficiency may be producible if a periodic material nonhomogeneity is included.

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